



## Effect of Simulated Coal-Derived Gas Composition on H<sub>2</sub>S Poisoning Behavior Evaluated Using a Disaggregation Scheme

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H<sub>2</sub>S poisoning is an important issue for solid oxide fuel cells (SOFCs) operated with syngas. The effect of simulated coal-derived gas composition on H<sub>2</sub>S poisoning behavior was evaluated using a disaggregation scheme where the influence of H<sub>2</sub> content was determined separately using a typical anode-supported SOFC operated with a N<sub>2</sub>/H<sub>2</sub> mixture gas, while the effect of other compositions (CO, CO<sub>2</sub>, and H<sub>2</sub>O) was investigated with simulated coal-derived gas having constant H<sub>2</sub> and CO flow rates balanced by a H<sub>2</sub>/N<sub>2</sub> mixture gas (83% H<sub>2</sub> and 17% N<sub>2</sub>). The results indicated that the extent of H<sub>2</sub>S poisoning was not pertinent to H<sub>2</sub> content when the cell was tested galvanostatically with a current density of 0.3 A/cm<sup>2</sup> at 800°C using a N<sub>2</sub>/H<sub>2</sub> mixture gas containing 10 ppm H<sub>2</sub>S, and the H<sub>2</sub>S poisoning impact can be completely removed by switching to sulfur-free gas. The CO, CO<sub>2</sub>, and high water vapor content aggravated the H<sub>2</sub>S poisoning effect, and the performance was almost irrecoverable when the cell was tested with a 35% H<sub>2</sub>-46% CO-16% N<sub>2</sub>-3% H<sub>2</sub>O mixture gas containing 12.5 ppm H<sub>2</sub>S. However, the introduction of 10% CO<sub>2</sub> and an increase in H<sub>2</sub>O content from 3 to 10% in the mixture gas can promote the performance recoverability to a larger extent.

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A wide variety of fuels, including coal gas, natural gas, biogas, and gasoline, can be used in a solid oxide fuel cell (SOFC) system even without pre-reforming. However, fuel impurities in coal syngas have been reported to deteriorate cell performance and shorten its life.<sup>1</sup> Because coal gas contains tens to thousands parts per million of H<sub>2</sub>S, and because a few parts per million of H<sub>2</sub>S in fuel can cause a great drop in cell power output, much attention has been paid to the behavior and mechanisms of H<sub>2</sub>S poisoning on SOFC.<sup>2-8</sup> Two different degradation mechanisms of H<sub>2</sub>S poisoning on Ni/yttria-stabilized zirconia (YSZ) cermets can be described as a sharp drop in cell performance caused by adsorption of S<sub>x</sub> on the surface of a nickel catalyst and a slow persistent degradation due to the formation of Ni<sub>x</sub>S<sub>y</sub>. Both mechanisms are strongly dependent on operating temperatures.<sup>9,10</sup>

Many investigations have been conducted on the poisoning behavior of H<sub>2</sub>S on the SOFC operated on pure H<sub>2</sub> or the mixture gas of H<sub>2</sub> and N<sub>2</sub>. Ishikura et al.<sup>2</sup> studied the influence of H<sub>2</sub>S concentration from 0.1 to 20 ppm on cell output voltage using H<sub>2</sub>/N<sub>2</sub> as fuel when the cell was operated under a constant current density of 200 mA/cm<sup>2</sup> at 900°C. It was shown that the output voltage dropped to zero as the cell was exposed to 20 ppm H<sub>2</sub>S and was not completely recovered after the removal of H<sub>2</sub>S. Rasmussen and Hagen<sup>3</sup> investigated the cell voltage drop caused by the addition of 2 to 100 ppm H<sub>2</sub>S in H<sub>2</sub> and suggested that 40 ppm H<sub>2</sub>S content was the saturation point.

Recently, some studies have been conducted focusing on the poisoning behaviors of H<sub>2</sub>S on the SOFC fueled by the coal-derived gas. Tremblay et al.<sup>8</sup> reported that the introduction of 200–240 ppm H<sub>2</sub>S led to a 10–12.5% degradation of cell performance with a simulated syngas of CO-H<sub>2</sub>-N<sub>2</sub>-H<sub>2</sub>O as fuel. Although the poisoning effect of H<sub>2</sub>S is indubitable, there are discrepancies on the poisoning extent of H<sub>2</sub>S on SOFC operated with different fuel gases. A 100% degradation was revealed using H<sub>2</sub>/N<sub>2</sub> containing 20 ppm H<sub>2</sub>S as fuel, while only 10–12.5% degradation can be detected when 200–240 ppm H<sub>2</sub>S was added in the syngas fuel composed of CO-H<sub>2</sub>-N<sub>2</sub>-H<sub>2</sub>O. Therefore, the composition of syngas may affect the sulfur poisoning extent on SOFC.

Previous work mainly focused on the poisoning effect of H<sub>2</sub>S on cell performance in terms of H<sub>2</sub>S concentration, operating temperatures, current density, and cell voltage,<sup>2,3</sup> but for an SOFC system operated with coal-derived gas containing H<sub>2</sub>S, it is necessary to

systematically diagnose the influence of the composition of coal syngas on H<sub>2</sub>S poisoning behaviors. This paper is thus initiated to investigate the effect of the coal syngas composition, including H<sub>2</sub>, N<sub>2</sub>, CO, CO<sub>2</sub>, and H<sub>2</sub>O, on the H<sub>2</sub>S poisoning behavior for typical anode-supported SOFCs.

### Experimental

Typical planar anode-supported 15 × 15 cm SOFCs were manufactured by batch at the Division of Fuel Cell and Energy Technology in Ningbo Institute of Material Technology and Engineering, Chinese Academy of Sciences. A 400 μm thick Ni/8YSZ anode substrate was prepared by tape casting, as were the other cell layers, including a 10 μm thick anode functional layer and a 10 μm thick 8YSZ electrolyte layer. The cathode of the cell was a double La<sub>0.8</sub>Sr<sub>0.2</sub>MnO<sub>3</sub> (LSM) layer sprayed on the anode substrate before cosintering. The overall cell dimensions for the test were 5 × 5.8 cm with an active area of 4 × 4 cm.

All of the cells were tested in an alumina testing house. Platinum and nickel foil were employed as current collectors at cathode and anode sides, respectively. Silver mesh sandwiched LSM was utilized for gas distribution at the cathode side where silver mesh was aimed to decrease the contact resistance between the cathode and the current collector.<sup>11</sup> For the anode side, two layers of nickel foam were applied for gas distribution and current collection.

Ceramic glass mainly consisting of silicon dioxide was used to achieve good sealing, and an extra 5.5 kg weight was applied to enhance the sealing effect. The cells were then placed into a furnace and heated in N<sub>2</sub> to 850°C with a heating rate of ~1°C/min to make the sealant intenerate. The reduction of the NiO-YSZ was performed under pure hydrogen with a flow rate of 0.3 standard liter per minute (SLM) for more than 5 h to ensure complete reduction of the anode. The furnace temperature was then decreased to 800°C, which was the testing temperature. The demonstration of the testing facility is shown in Fig. 1.

After the anode was completely reduced, the air was used as a cathode oxidant with a flow rate of 2 SLM, while pure hydrogen was introduced into the anode side as fuel gas with a flow rate of 0.5 SLM. The cell was discharged galvanostatically at 0.3 A/cm<sup>2</sup> for at least 1 h to ensure sharp electrode activation.<sup>12,13</sup> The current-voltage (*I*-*V*) curves were recorded at different mixing ratios of H<sub>2</sub>/N<sub>2</sub> as anode gas (H<sub>2</sub> content from 27 to 53%). A 10 ppm H<sub>2</sub>S balanced by N<sub>2</sub> was subsequently injected into the cell system fueled with the above H<sub>2</sub>/N<sub>2</sub> mixture gas for 60 min and then removed. The degradation in cell performance was recorded in terms of voltage drop at a constant discharging current density of 0.3 A/cm<sup>2</sup>. The electrochemical impedance spectra (EIS) were ob-

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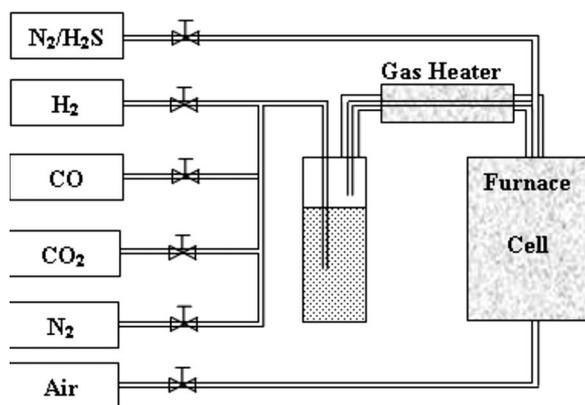


Figure 1. Illustration of the testing facility.

tained at open-circuit voltage (OCV) by four-electrode configuration using an electrochemical workstation (IM6ex, ZAHNER) with a scanning frequency range from 0.1 to 2 MHz before the 10 ppm  $\text{H}_2\text{S}$  injection and 30 min after  $\text{H}_2\text{S}$  removal, respectively. The impedance spectra in galvanostatic mode were also recorded at a current density of  $0.3 \text{ A/cm}^2$  at  $800^\circ\text{C}$  for further inspection.

In exploring the effect of the compositions ( $\text{CO}$ ,  $\text{CO}_2$ , and  $\text{H}_2\text{O}$ ) in simulated coal-derived gas on  $\text{H}_2\text{S}$  poisoning behavior, a  $\text{H}_2/\text{N}_2$  mixture gas (83%  $\text{H}_2$  and 17%  $\text{N}_2$ ) was used as the initial base fuel, and  $\text{CO}$ ,  $\text{H}_2\text{O}$ , and  $\text{CO}_2$  were added into the base fuel to form mixture gases of 35%  $\text{H}_2$ –46%  $\text{CO}$ –16%  $\text{N}_2$ –3%  $\text{H}_2\text{O}$ , 31%  $\text{H}_2$ –42%  $\text{CO}$ –12%  $\text{CO}_2$ –12%  $\text{N}_2$ –3%  $\text{H}_2\text{O}$ , and 29%  $\text{H}_2$ –38%  $\text{CO}$ –10%  $\text{CO}_2$ –13%  $\text{N}_2$ –10%  $\text{H}_2\text{O}$ , where  $\text{H}_2$  and  $\text{CO}$  flows were set as 0.26 and 0.265 SLM, respectively, throughout the experiments for all the mixture gases. The cell was tested under these mixture gases with the injection of 12.5 ppm  $\text{H}_2\text{S}$  for 180 min at a current density of  $0.25 \text{ A/cm}^2$ . Impedance spectra were recorded at OCV after removing  $\text{H}_2\text{S}$  for 10 and 660 min, respectively. For comparison, the impedance spectra were also recorded before poisoning. The durability of the cell operated with a simulated coal-derived gas of 29%  $\text{H}_2$ –38%  $\text{CO}$ –10%  $\text{CO}_2$ –13%  $\text{N}_2$ –10%  $\text{H}_2\text{O}$  was also investigated up to 10 h at a current density of  $0.25 \text{ A/cm}^2$ .

### Results and Discussion

Investigations were conducted to explore the influence of each composition in simulated coal-derived gas containing  $\text{H}_2$ ,  $\text{CO}$ ,  $\text{CO}_2$ ,  $\text{H}_2\text{O}$ , and  $\text{N}_2$  on  $\text{H}_2\text{S}$  poisoning behavior. The  $\text{H}_2$  content in  $\text{H}_2/\text{N}_2$  mixture gas was thus evaluated first, and the effect of the addition of  $\text{CO}$ ,  $\text{CO}_2$ , and  $\text{H}_2\text{O}$  was subsequently determined.

**Effect of  $\text{H}_2$  content on  $\text{H}_2\text{S}$  poisoning behavior.**— Figure 2 shows  $I$ - $V$  curves of a single cell fueled with  $\text{H}_2/\text{N}_2$  mixture gas after the cell is completely reduced for 5 h. It can be seen that the performance of the cell operated at  $800^\circ\text{C}$  is moderate. The maximum power output ( $P_{\text{max}}$ ) increases from 287 to  $385 \text{ mW/cm}^2$  with the increase  $\text{H}_2$  content from 27% (0.4 SLM) to 53% (0.8 SLM), while the flow rate for  $\text{N}_2$  remains constant. The OCV for each composition of  $\text{H}_2/\text{N}_2$  is almost the same, reaching a potential value of about 1.1 V.

Figure 3a shows the degradation in cell voltage by exposure to 10 ppm  $\text{H}_2\text{S}$  when the cell is fueled with various ratios of  $\text{H}_2/\text{N}_2$  at a current density of  $0.3 \text{ A/cm}^2$ . The cell voltage drops about 3.2–4.2% from the initial voltages of 761–780 mV by 10 ppm  $\text{H}_2\text{S}$  injection for 60 min. and after switching to sulfur-free gas, the cell voltage can almost be fully recovered within 30 min. The output voltage exhibits a small abrupt drop once  $\text{H}_2\text{S}$  is introduced for all five gas compositions and then gradually reaches a stable value after 35 min. When stopping  $\text{H}_2\text{S}$  flow, a small abrupt recovery is observed, and the voltage then immediately returns to its initial value within 25 min. Figure 3b shows the EIS measured at OCV before

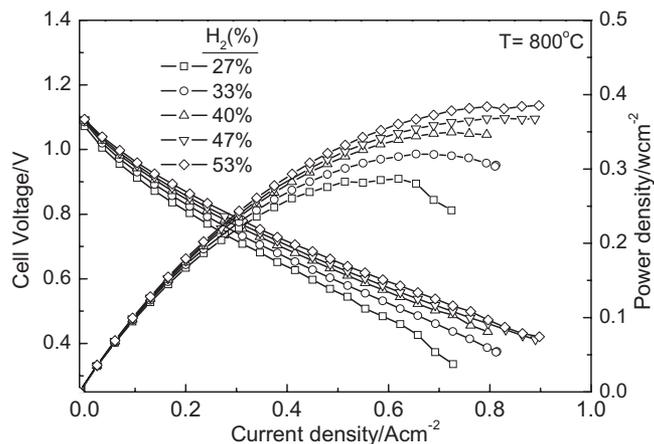


Figure 2.  $I$ - $V$  curves measured using different ratios of  $\text{H}_2/\text{N}_2$  at  $800^\circ\text{C}$ .

$\text{H}_2\text{S}$  introduction and after  $\text{H}_2\text{S}$  removal for 30 min. The results obtained before  $\text{H}_2\text{S}$  injection show that the polarization resistance ( $R_p$ ) increases with the increase in  $\text{N}_2$  content, which results in relatively low cell performance. After removing  $\text{H}_2\text{S}$  injection for 30 min,  $R_p$  can almost go back to its original value, which agrees with the results from Fig. 4a. Comparing the results before poisoning with the results after stopping  $\text{H}_2\text{S}$  injection for 30 min, no legible deviation was detected for all five  $\text{H}_2$  content compositions, indicat-

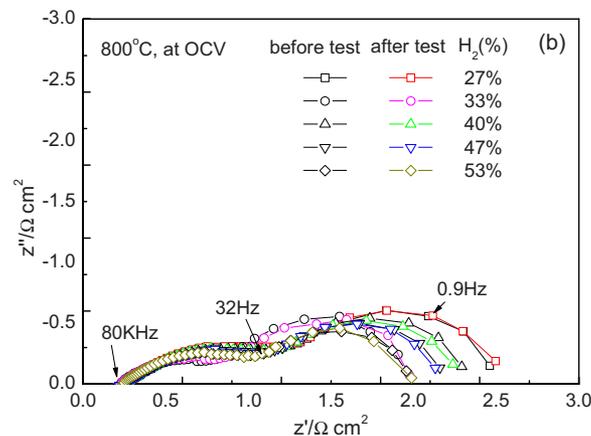
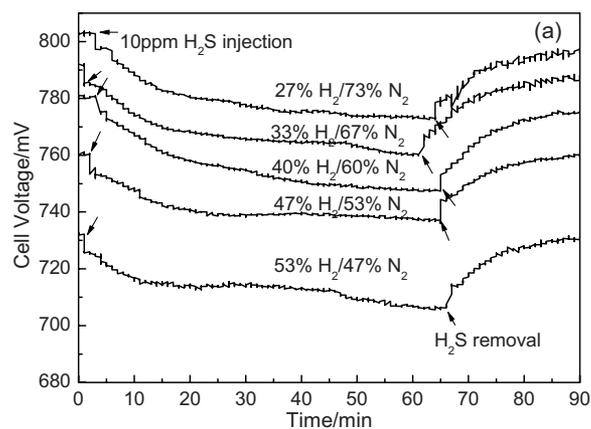
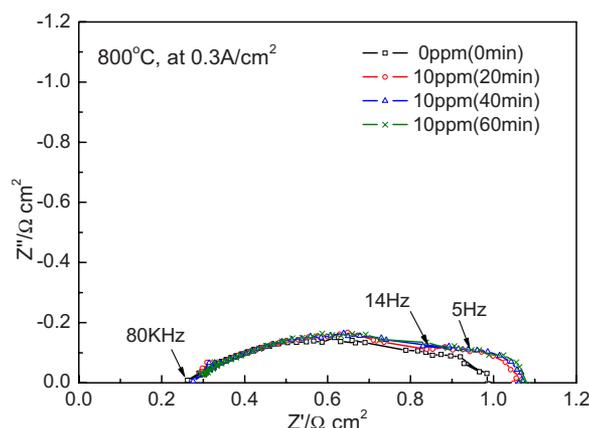


Figure 3. (Color online) (a) The variation in output voltage for the single cell operated with  $\text{H}_2/\text{N}_2$  mixture gas at a current density of  $0.3 \text{ A/cm}^2$  and a temperature of  $800^\circ\text{C}$ . (b) Impedance spectra at OCV before 10 ppm  $\text{H}_2\text{S}$  injection and removal for 30 min.

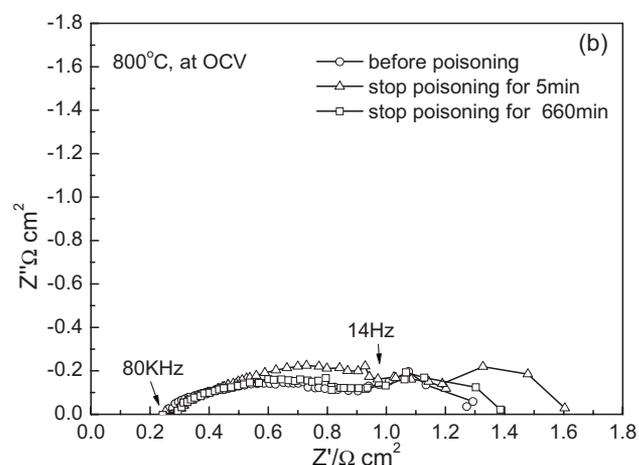
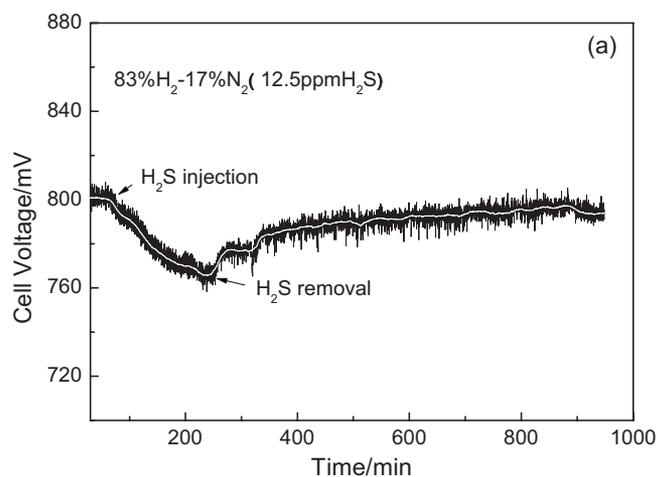


**Figure 4.** (Color online) Impedance spectra curves for the cell fueled with 83%  $N_2/17\% H_2$  containing 12.5 ppm  $H_2S$  obtained in a galvanostatic mode after  $H_2S$  injection for 0, 20, 40, and 60 min, respectively.

ing the recoverability of the  $H_2S$ -poisoned cell by switching to sulfur-free gas as reported.<sup>2</sup> The results of impedance spectra show that the  $R_p$  increases after injecting  $H_2S$ , while  $R_s$  remains constant, suggesting that introducing  $H_2S$  to fuel can increase the difficulty in some electrochemical processes occurring in the anode side.

The cell voltage drop caused by 10 ppm  $H_2S$  is listed in Table I where the cell is operated with  $H_2/N_2$  mixture gases of different ratios. The largest drop in cell voltage is  $\sim 4.2\%$  from 780 mV and occurs when the cell is operated with 40%  $H_2/60\% N_2$ , while the smallest degradation of 3.2% from 761 mV is obtained with the application of 33%  $H_2/67\% N_2$ . Decreasing  $H_2$  content in the fuel gas cannot decrease the  $H_2S$  poisoning effect. This is different from the results reported by Lohsoontorn et al.<sup>4</sup> where polarization resistance and bulk resistance increase with the decrease in  $H_2$  content from 97 to 9.7%. But nevertheless, no evident difference in voltage drop is observed in this study when decreasing  $H_2$  content. Sulfur compound formed at the anode side may be immediately reduced back to sulfur hydrogen in a  $H_2S$  free fuel environment at least by thermodynamic consideration. This reduction process is dependent on  $p_{H_2O}/p_{H_2}$ .<sup>14</sup> The results in this paper indicate that a 27%  $H_2$  content in  $N_2/H_2$  mixture fuel gas is sufficient to supply a reducing atmosphere to reduce the  $H_2S$  poisoning effect. Moreover, the initial voltage ( $U_0$ ) increases with increasing  $H_2$  content and has no significant influence on the  $H_2S$  poisoning effect, which may also be ascribed to the saturation of  $H_2$  content in this study.

To explore the mechanism of the  $H_2S$  poisoning cell, impedance spectra were monitored and recorded, as shown in Fig. 4, where the cell was fueled with 33%  $H_2-67\% N_2$  at 0.3 A/cm<sup>2</sup> after injecting  $H_2S$  for 0, 20, 40, and 60 min. As shown in Fig. 4,  $R_p$  increases rapidly once 10 ppm  $H_2S$  is introduced within 20 min and then gradually reaches an equilibrium value until exposure to  $H_2S$  for 60 min. This agrees with the results of the electrical properties, as shown in Fig. 3a. The difference in impedance spectra exists in a frequency zone ranging from 0.1 to 5 Hz (14 Hz), which is reported



**Figure 5.** (a) Effect of  $CO$  addition to the mixture gas on 12.5 ppm  $H_2S$  poisoning behavior and (b) impedance spectra at OCV before and after poisoning.

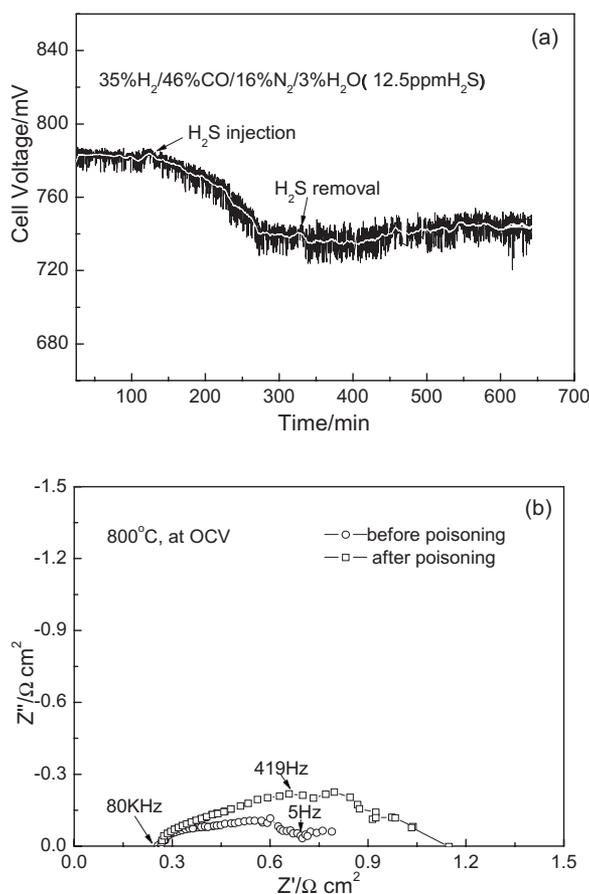
to involve diffusion and gas conversion at the anode.<sup>15</sup> This indicates that the presence of  $H_2S$  can block gas diffusion and bring hindrance in gas conversion at the anode side, thereby leading to an immediate drop in cell performance with the injection of  $H_2S$ . However, the microstructural inspection using scanning electron microscope shows that the anode microstructure after poisoning tests is almost the same as the anode without injection of  $H_2S$  even for the cell exhibiting highest degradation by  $H_2S$  poisoning. The lack of microstructural evidence may be due to short-term poisoning and the small amount of  $H_2S$  (10 ppm).<sup>10</sup>

*Effect of other compositions of simulated coal-derived syngas.*—Figure 5a shows the variation in cell voltage caused by exposure to 12.5 ppm  $H_2S$  with a constant discharging current den-

**Table I.** The drop extent of cell voltage caused by 10 ppm  $H_2S$  when operated with different ratios of  $N_2-H_2$  mixture gas at a current density of 0.3 A/cm<sup>2</sup> and a temperature of 800°C.

	27% $H_2/73\% N_2$	33% $H_2/67\% N_2$	40% $H_2/60\% N_2$	47% $H_2/53\% N_2$	53% $H_2/47\% N_2$
$U_0$ /mV	732	761	780	792	803
$U_{end}$ /mV	705	737	747	760	772
$\Delta U$ (%)	-3.7%	-3.2%	-4.2%	-4.0%	-3.9%
$U_r^a$ (%)	100%	100%	99.6%	99.4%	99.3%

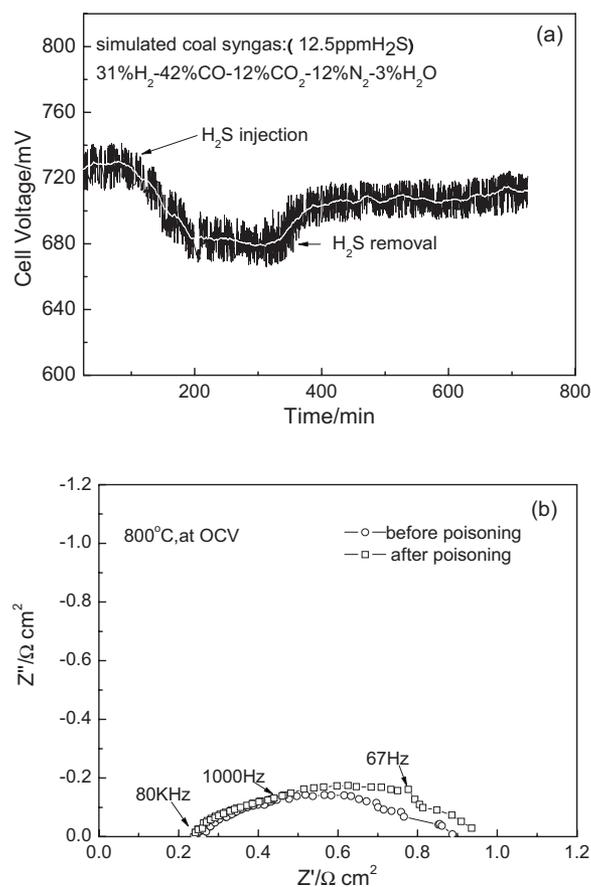
<sup>a</sup>  $U_r$  denotes the recovery percentage of the voltage.



**Figure 6.** (a) Effect of the CO<sub>2</sub> addition to the mixture gas on 12.5 ppm H<sub>2</sub>S poisoning behavior and (b) impedance spectra at OCV before and after poisoning.

sity of 0.25 A/cm<sup>2</sup> when the cell is fueled by 83% H<sub>2</sub> and 17% N<sub>2</sub>. The voltage drop is 35 mV, equivalent to 4.4% of the initial voltage 805 mV. However, no abrupt drop is observed with the addition of H<sub>2</sub>S, which is different from the poisoning behavior discussed above. The voltage tends to continue decreasing without reaching a stable value within 3 h, and the total degradation in the cell is more serious by comparison with the results in Fig. 3a. This may be attributed to the change of silver mesh sandwiched nickel foil to nickel foam as current collector at the anode side because other testing conditions in this test are identical to the previous tests. However, the mechanism for this difference in degradation tendency is not clear. One explanation may be that using nickel foam as the anode for current collecting can suppress a sharp drop in cell performance, thereby demonstrating a persistent gradual degradation. In addition, the voltage is almost completely recovered as soon as sulfur-free gas is flowed in, giving evidence of less poisoning impact with the use of nickel foam. The results derived from impedance spectra are given in Fig. 5b where polarization resistance continues to decrease after stopping H<sub>2</sub>S flow for 5–660 min, and the deviation of frequency range is below 14 Hz, indicating that the H<sub>2</sub>S poisoning leads to difficulties in some electrochemical processes in the anode; these difficulties can be slowly eliminated once the H<sub>2</sub>S is removed.

Figure 6a displays the influence of the introduction of CO into fuel (31% H<sub>2</sub> and 16% N<sub>2</sub>) on the H<sub>2</sub>S deteriorating effect. The tendency of the initial voltage drop is the same as the cell operating with 83% H<sub>2</sub> and 17% N<sub>2</sub>. The degradation percentage reaches 5%, ~45 mV, slightly higher than that using H<sub>2</sub>/N<sub>2</sub> as fuel gas (4.4%). However, after removal of H<sub>2</sub>S for 300 min, only little recovery is achieved. This indicates that the poisoning for the cell tested in



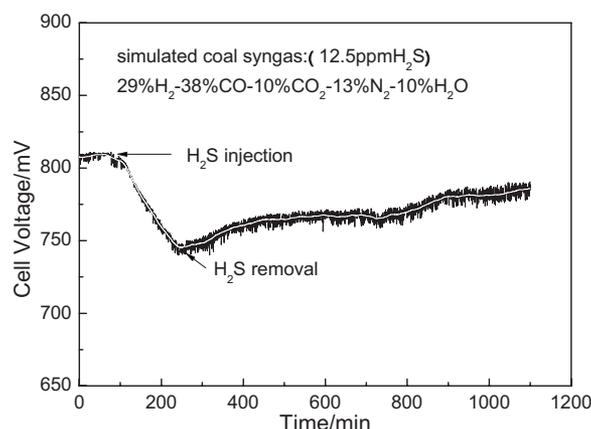
**Figure 7.** (a) Effect of increasing H<sub>2</sub>O content on 12.5 ppm H<sub>2</sub>S poisoning behavior and (b) impedance spectra at OCV before and after poisoning.

H<sub>2</sub>-CO-N<sub>2</sub> cannot be completely removed after stopping H<sub>2</sub>S flow, which is further confirmed in Fig. 6b by the results of impedance spectra measured before H<sub>2</sub>S poisoning and after stopping H<sub>2</sub>S flow for 300 min. The second and third arcs in the impedance spectra increase greatly, representing that the charge transfer between Ni and YSZ is also detrimentally affected.<sup>15,16</sup> This may be attributed to the adsorption of S on the surface of Ni or the formation of Ni<sub>x</sub>S<sub>y</sub>.<sup>9,10,17</sup>

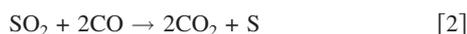
The H<sub>2</sub>S poisoning behavior after the addition of 12% CO<sub>2</sub> into the above mixture gas (H<sub>2</sub>-CO-N<sub>2</sub>) is shown in Fig. 7a. The cell voltage also exhibits a gradual degradation mode, which is identical to the above two tests. However, the degradation percentage increases to nearly 6.9% at ~53 mV, about 1.9% higher than that of the cell operated with H<sub>2</sub>-CO-N<sub>2</sub>. The voltage is detected to be recovered to about 97% of the initial voltage value after H<sub>2</sub>S removal. This indicates that the addition of CO<sub>2</sub> can be beneficial to mitigate the degradation caused by H<sub>2</sub>S poisoning. Figure 7b shows the impedance spectra curves obtained before H<sub>2</sub>S poisoning and after stopping the H<sub>2</sub>S flow when the single cell is fueled with the above H<sub>2</sub>-CO-N<sub>2</sub>-CO<sub>2</sub> mixture gas. As shown in Fig. 7b, the incomplete recovery of single cell performance is further proved by impedance spectra. The polarization resistance slightly increases after poisoning, whereas the increasing extent is somewhat smaller by comparison with the value obtained from Fig. 6b.

As shown in Eq. 1, H<sub>2</sub>S can be electrochemically oxidized to SO<sub>2</sub> by oxygen.<sup>17</sup> Because SO<sub>2</sub> may be present at the cell anode, it would react with CO addition to form CO<sub>2</sub> and sulfur (Eq. 2) resulting in degradation to a larger extent and irrecoverable performance drop





**Figure 8.** Voltage vs operation time for the single cell operated with a 29% H<sub>2</sub>-38% CO-10% CO<sub>2</sub>-13% N<sub>2</sub>-10% H<sub>2</sub>O simulated coal gas containing 12.5 ppm H<sub>2</sub>S.



In the mixture gas, a reversible chemical reaction involving CO<sub>2</sub> can be described as Eq. 3 at the operation temperatures of SOFC.<sup>18,19</sup> The addition of CO<sub>2</sub>



can actually increase the CO and water content in fuel gas, leading to a twofold effect on the testing result: (i) Enhancement of CO leads to more deterioration in cell performance, as explained by Eq. 2 and (ii) water product may convert S<sub>n</sub> to H<sub>2</sub>S and SO<sub>2</sub>, as shown in Eq. 4.<sup>17</sup> The production of H<sub>2</sub>S and SO<sub>2</sub>

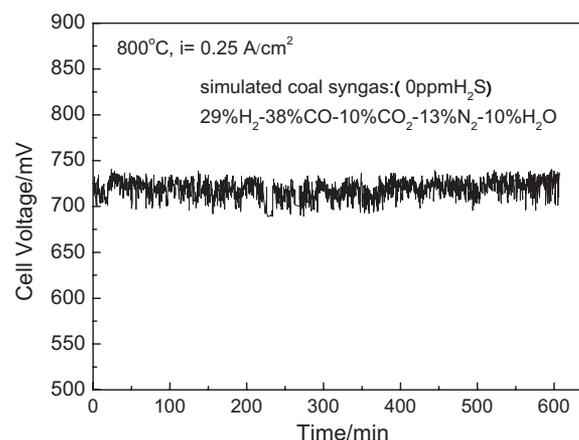


may be partially blown out from the anode gas chamber or electrochemical active zone and thus lead to a partial performance recovery.

To investigate the effect of water content in coal-derived gas on the H<sub>2</sub>S poisoning behavior, the vapor content is increased from 3 to 10%,<sup>1</sup> and thus the fuel composition is determined as 29% H<sub>2</sub>-38% CO-10% CO<sub>2</sub>-13% N<sub>2</sub>-10% H<sub>2</sub>O. Figure 8 shows the variation in voltage with time for the cell fueled with 29% H<sub>2</sub>-38% CO-10% CO<sub>2</sub>-13% N<sub>2</sub>-10% H<sub>2</sub>O containing 12.5 ppm H<sub>2</sub>S. The output voltage for the cell operated with the fuel containing H<sub>2</sub>S is degraded to 7.8% at ~63 mV, 0.9% higher than that of the cell operated with 31% H<sub>2</sub>-42% CO-12% CO<sub>2</sub>-2% N<sub>2</sub>-3% H<sub>2</sub>O in Fig. 7a where the fuel has similar components except for the amount of water vapor. After H<sub>2</sub>S removal for 800 min, the output voltage recovers to about 97% of the initial value and becomes stable thereafter. It seems that enhancing water content results in a more serious voltage drop, but it is also more helpful to the performance recovery of the single cell fueled with coal-derived gas. As interpreted by Eq. 4, water may hold back the formation of sulfur by the reaction of H<sub>2</sub>S and SO<sub>2</sub>, thereby permitting a larger performance recovery. At the same time, sulfur can also be produced by H<sub>2</sub>S decomposition (Eq. 5)



so that the adsorption of S<sub>2</sub> on the Ni surface may result in an immediate drop in cell voltage. If more S adsorption occurs directly and immediately on the nickel catalyst surface, the S may react with Ni to form Ni<sub>3</sub>S<sub>2</sub>,<sup>10</sup> leading to a larger performance recovery. The results in Fig. 8 are thus compared with the variation in voltage with operation time for the single cell using the same fuel, 29% H<sub>2</sub>-38% CO-10% CO<sub>2</sub>-13% N<sub>2</sub>-10% H<sub>2</sub>O, without H<sub>2</sub>S at the same current density, as shown in Fig. 9. Apparently, the cell not exposed to H<sub>2</sub>S is operated steadily with the above simulated



**Figure 9.** Cell voltage vs operation time for the single cell operated at a current density of 0.25 A/cm<sup>2</sup> and 800°C using a 29% H<sub>2</sub>-38% CO-10% CO<sub>2</sub>-13% N<sub>2</sub>-10% H<sub>2</sub>O simulated syngas without H<sub>2</sub>S.

syngas for 600 min without any degradation at 800°C, suggesting that only H<sub>2</sub>S causes the cell performance variation.

## Conclusions

Investigations have been conducted on the effect of H<sub>2</sub> content in H<sub>2</sub>/N<sub>2</sub> mixture gas and the effect of other compositions in simulated coal-derived gas on H<sub>2</sub>S poisoning behavior. Increasing the H<sub>2</sub> content from 27 to 53% has no influence on the H<sub>2</sub>S poisoning effect. The degradation impact caused by exposure to H<sub>2</sub>S is completely removed after H<sub>2</sub>S removal when the cell is fueled with H<sub>2</sub>/N<sub>2</sub>. When the cell is operated with simulated coal syngas, the addition of CO, CO<sub>2</sub>, and H<sub>2</sub>O aggravates the H<sub>2</sub>S poisoning effect; however, increasing the CO<sub>2</sub> and H<sub>2</sub>O content may be helpful for performance recovery.

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## References

1. F. N. Cayan, M. J. Zhia, S. R. Pakalapatia, I. Celika, N. Q. Wua, and R. Gemmenb, *J. Power Sources*, **185**, 595 (2008).
2. A. Ishikura, S. Sakuno, N. Komiya, H. Sasatsu, N. Masuyama, H. Itoh, and K. Yasumoto, *ECS Trans.*, **7**(1), 845 (2007).
3. J. F. B. Rasmussen and A. Hagen, *J. Power Sources*, **191**, 534 (2009).
4. P. Lohsoontorn, D. J. L. Brett, and N. P. Brandon, *J. Power Sources*, **183**, 232 (2008).
5. S. W. Zha, Z. Cheng, and M. L. Liu, *J. Electrochem. Soc.*, **154**, B201 (2007).
6. K. Sasaki, K. Susuki, A. Iyoshi, M. Uchimur, N. Imamura, H. Kusaba, Y. Teraoka, H. Fuchino, K. Tsujimoto, Y. Uchida, et al., *J. Electrochem. Soc.*, **153**, A2023 (2006).
7. K. Sasaki, S. Adachi, K. Haga, M. Uchikawa, J. Yamamoto, A. Iyoshi, J.-T. Chou, Y. Shiratori, and K. Itoh, *ECS Trans.*, **7**(1), 1675 (2007).
8. J. P. Tremblay, A. I. Marquez, T. R. Ohrn, and D. J. Bayless, *J. Power Sources*, **158**, 263 (2006).
9. Z. Cheng and M. L. Liu, *Solid State Ionics*, **178**, 925 (2007).
10. J. Dong, Z. Cheng, S. W. Zha, and M. L. Liu, *J. Power Sources*, **156**, 461 (2006).
11. S. P. Jiang, J. G. Love, and L. Apateanu, *Solid State Ionics*, **160**, 15 (2003).
12. V. A. C. Haanappel, A. Mai, and J. Mertens, *Solid State Ionics*, **177**, 2033 (2006).
13. S. Koch, M. Mogensen, P. V. Hendriksen, N. Dekker, and B. Rietveld, in *Proceedings of the Sixth European Solid Oxide Fuel Cell Forum*, European Fuel Cell

- Forum, p. 1000, Lucerne, Switzerland (2004).
14. Z. Cheng, S. W. Zha, and M. L. Liu, *J. Power Sources*, **172**, 688 (2007).
  15. R. Barfod, A. Hagen, S. Ramousse, P. V. Hendriksen, and M. Mogensen, *Fuel Cells*, **6**, 141 (2006).
  16. R. Barfod, M. Mogensen, T. Klemensø, A. Hagen, Y. L. Liu, and P. Vang Hendriksen, *J. Electrochem. Soc.*, **154**, B371 (2007).
  17. A. Lussier, S. Sofie, J. Dvorak, and Y. U. Idzerda, *Int. J. Hydrogen Energy*, **33**, 3945 (2008).
  18. J. N. Kuhn, N. Lakshminarayanan, and U. S. Ozkan, *J. Mol. Catal. A: Chem.*, **282**, 9 (2008).
  19. F. P. Nagel, T. J. Schildhauer, J. Sfeir, A. Schuler, and S. M. A. Biollaz, *J. Power Sources*, **189**, 1127 (2009).